

## Calcination, Reduction and Sintering of ADU Spheres for HTGR Fuel

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### 1. Introduction

The international oil market is again in turmoil in accordance with the increasing of human needs and energy consumption. Soaring oil prices, fears of supply security, and climate change are concerned becoming more concrete make for an uncertain energy future. In this view point, nuclear energy is an important, yet controversial option for energy supply.

High Temperature Gas Reactor will play a dominant role in the worldwide fleet of nuclear reactors of the next decade for electricity production and high temperature heat [1]. HTGR have two reactor types which use the basic fuel concept based on the dispersion of TRISO coated particles in graphite in shown Fig.1.

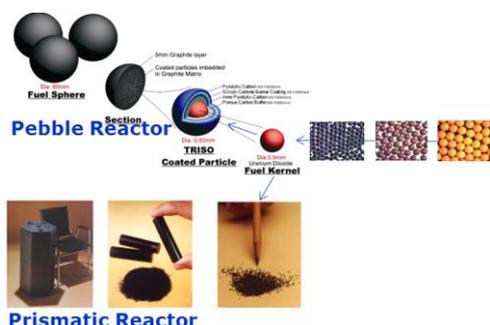


Figure 1. General concepts of VHTR fuel.

The TRISO coated particle for these purposes is prepared with pyro-carbon and silicone carbide coatings on a spherical  $UO_2$  kernel surface as fissile material. The TRISO fuel particle consists of a microsphere (i.e.,  $UO_2$  kernel) of nuclear material; encapsulated by multiple layers of pyro-carbon and a SiC layer [2,3].

This multiple coating layers system has been engineered to retain the fission products generated by fission of the nuclear material in the kernel during normal operation and all licensing basis events over the design lifetime of the fuel.

$UO_2$  kernels are produced by using the modified sol-gel process, a wet process, generally known as the GSP method. Wet chemical processes are flexible in producing kernels of different size and chemical composition with high throughput and yield, good spherical shape, and narrow size distribution. This chemical processing route is well-known to the potential kernel fabrication processes [4].

The principle, as set out in Fig.2, involves first of all preparing a pseudo-sol(also known as a "broth") from an initial uranyl nitrate solution. This broth solution is obtained through addition of a number of additives, as determined by process know-how, including a soluble

organic polymer, that are subsequently gels into droplets and are dispersed for ADU precipitation.

The formation of spheres is then formed by the dispersed in air and ammonia gases atmosphere.

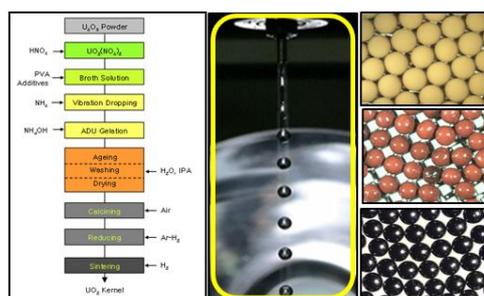


Figure 2. Schematic material flow diagram (left) and particle shapes at each step.

The gelation of polymer was induced at the surface of the droplets in this step. The surfaces of these liquid droplets are slightly solidified by reaction of ammonia gas and uranyl ion into the droplets. Spherical droplets are converted to aged-ADU gels in  $NH_4OH$  solution. Then, many steps, such as the ageing and drying of ADU gel, the calcining to  $UO_3$ , the sintering to  $UO_2$ , were progressed continuously.

### 2. $UO_2$ kernel preparation

The experimental apparatus for the spherical  $UO_2$  kernel mainly consisted of a broth solution preparation system, vibrating dropping system, AWD(Ageing-Washing-Drying) system, and thermal treatment system. The shape and size of the ADU gel and  $UO_3$  particles, and inside microstructure of a final  $UO_2$  kernel were observed by means of a SEM.

### 3. Results

$UO_2$  kernel prepared with the optimum experimental conditions using the above explained procedures. Here, the sintered- $UO_2$  spheres were obtained after sintering of calcined- $UO_3$  spheres. The specimen, shown in Figure 3, was obtained from the  $UO_2$  spheres made by resin treating procedure for the observation of the inside microstructure  $UO_2$  sphere. Figure 3 showed the SEM photos at a horizontal section of a final sintered- $UO_2$  kernel. In calcining process of dried-ADU gel particles, the heating rates were changed from  $0.5^\circ C/min$  to  $5^\circ C/min$ . with an interval of  $1\sim 2^\circ C/min$ .

Microstructures at the cross section of  $UO_2$  kernel were obtained at low heating rate, from  $0.5^\circ C/min$ . to  $3^\circ C/min$ ., in calcination process were comparatively good states, as shown in at (a) and (b) of Figure 3, but in

the case of (c), the inside microstructure was not better than (a) or (b). Small space containing the macro pore and micro pore existed in center of the  $\text{UO}_2$  kernel.

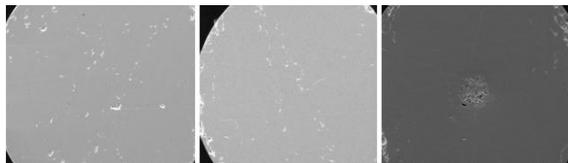


Figure 3.  $\text{UO}_2$  inside microstructure according to the change of heating rate in calcining step, (a) 1 °C/min, (b) 2 °C/min, and (c) 5 °C/min.

If the heating rate is faster, it means that is a high heating rate, is critical value in calcining step, the thermal treated- $\text{UO}_3$  particles were cracked or broken, or the shrinkage to reach the center of kernel was not occurred as shown in Figure 4.

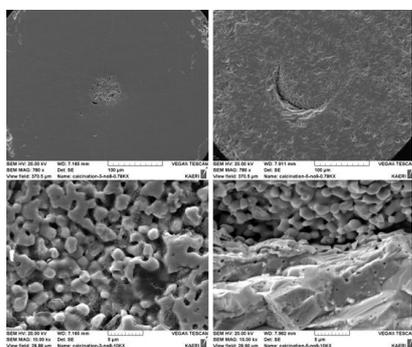


Figure 4. The microstructures of  $\text{UO}_2$  kernel center cross section, in the case of a fast heating rate in the calcining step.

Otherwise, the cross section shape of a final sintered  $\text{UO}_2$  sphere was obtained from calcined- $\text{UO}_3$  in air condition of 450°C. Figure 5 shows the sample specimen obtained from polishing of a spherical  $\text{UO}_2$  particle with height. Inside microstructures of three samples were well state at near top, middle, and center of kernel.

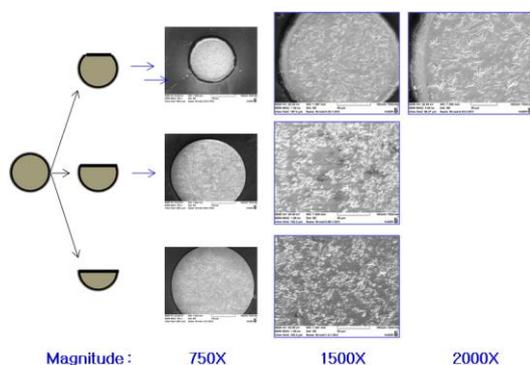


Figure 5. SEM photos on microstructure at cross section of an  $\text{UO}_2$  kernel.

The state of microstructure appeared same shapes and crystallite sizes. Therefore, the heating rate at the calcining step should be kept to below 3 °C/min. in preparing of spherical  $\text{UO}_2$  kernel.

#### 4. Conclusion

In this study, the technical reviews and laboratory scale experiments were carried out about the preparation of ADU gel particles and the thermal treatments on  $\text{UO}_3$  calcination and  $\text{UO}_2$  sintering processes to obtain a spherical  $\text{UO}_2$  kernel. The most important factors are the mole ratio of uranium concentration and organic additives and the heating rate profiles in the calcining and the sintering processes.

#### REFERENCES

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