

## Non-Rod Experiments for Improving Atomization Technology and $U_3Si_2$ Powder Fabrication at KAERI

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### 1. Introduction

Korea Atomic Energy Research Institute(KAERI) developed the atomization technology for U-Si powder for HANARO research reactor fuel and successfully fabricated the atomized U-Si powder in 1991[1]. Since then, KAERI has been developing atomization technology presented in Fig. 1.

Atomized powder is known for its high purity with fewer defects and high production yield rates, because the atomization process is much simpler than the conventional crushing methods. Compared to the conventional pulverized powder fabricated by the crushing methods, the atomized powder, which is spherical in shape, has advantages such as high uranium loading capability, various U alloy compositions, and homogenous particle distribution[2].

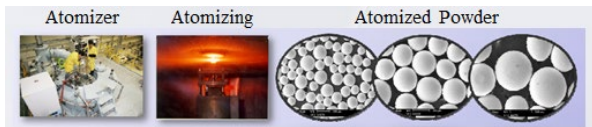


Fig. 1. KAERI centrifugal atomization technology [3]

Coating agents are applied between the pouring rod and the ceramic crucible shown in Fig. 2 (Extracting Part of the Drawing) to prevent the molten U alloy from leaking. When the target temperature is reached, the pouring rod is lifted so that the molten U alloy flows down through the nozzle of the ceramic crucible by gravity. The molten U alloy then reaches a rotating disk and is sprayed as droplets. These droplets cool rapidly and form fine spherical particles with diameters less than  $150\ \mu m$ .

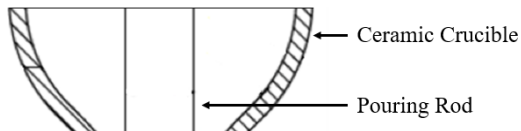


Fig. 2. Ceramic Crucible and Pouring Rod

For  $U_3Si_2$  atomization, the molten  $U_3Si_2$  alloy must be heated to approximately  $2000^\circ C$  as the target temperature. This severe thermal condition can cause significant damage on the components such as the pouring rod and the coating agents, leading to low yield rates or failures.[4]

Under those  $U_3Si_2$  atomization process conditions, leak failures related to high temperatures near  $2000^\circ C$  of the molten  $U_3Si_2$  alloy occurred, causing the molten  $U_3Si_2$  alloy to dribble down onto the disk without lifting the rod. To address this issue, the pouring rod was removed so that coating agents were no longer needed between the rod and the crucible.

The properties of the coating agents, which indicated leak failures occurring near  $2000^\circ C$ , were applied to the nozzle of the crucible instead. The goal of this non-rod experiment was to check and ensure that the molten  $U_3Si_2$  alloy flowed down onto the disk automatically without the rod after reaching  $2000^\circ C$ .

### 2. Methods and Result

The non-rod experiment's conditions and process parameters could not be fully described due to the confidentiality for KAERI's atomization technology, which has been designated as a National Core Technology by the Ministry of Trade, Industry and Energy in Korea.

The non-rod experiments were conducted four times using two different types of coating methods. The same method was applied to each pair of experiments; the same process conditions for loadings and temperatures were applied throughout all experiments, as detailed in Table I.

Table I: Non-rod Experiment's Conditions for  $U_3Si_2$  Atomized Powder

Batch Name	Applying Coating Agents on Nozzle	Loading (g)	Temp. ( $^\circ C$ )
C2405	1 <sup>st</sup> Method	<500	<2100
C2406	1 <sup>st</sup> Method	<500	<2100
C2407	2 <sup>nd</sup> Method	<500	<2100
C2408	2 <sup>nd</sup> Method	<500	<2100

In the first experiment (C2405 batch), the primary leak occurred at  $1150^\circ C$  when U began to melt and started to form an alloy with Si. A secondary leak occurred at  $1900^\circ C$ , which appeared to be related to instability in the coating agents. At  $2050^\circ C$ , leakage of

the entire molten  $U_3Si_2$  alloy began, but instead of trickling at the center of the disk, the molten alloy trickled onto the edges. As shown in Figure 3, this resulted in an unstable powder spraying process and some molten alloy leaked through cracks in the exterior of the ceramic crucible, rather than through the nozzle, and was found to have adhered to the graphite components. After the sieving process, the yield for  $U_3Si_2$  powder with diameters less than  $150\ \mu m$  was 15.8%.



Fig. 3. Atomization Component Parts after first experiment (C2405 batch) : Disk(Left), Carbon Parts(Right)

In the second experiment (C2406 batch), the same process conditions as the first experiment were repeated. At  $1150^\circ C$ , where U and Si started to react chemically, the leakage of the entire molten  $U_3Si_2$  alloy occurred. Since there were no traces of the molten alloy on the exterior of the ceramic crucible or near the graphite components, it suggested that the molten  $U_3Si_2$  alloy leaked only through the nozzle and solidified in the ingot as shown in Figure 4. The leakage problem that occurred early stage before reaching  $2000^\circ C$  was caused by the failure to apply the coating agents properly, resulting in 0% yield rate of  $U_3Si_2$  powder.



Fig. 4. Atomization Component Parts after second experiment (C2406 batch) : Disk and Ingot(Left), Carbon Parts(Right)

In the third experiment (C2407 batch), the issues observed in the previous experiments, such as leaks at  $1150^\circ C$  and  $1900^\circ C$ , were resolved. Consequently, the temperature was successfully increased to  $2050^\circ C$  without any leaks. At  $2050^\circ C$ , the molten  $U_3Si_2$  alloy automatically leaked and trickled out onto the disk. Subsequently, Ar gas was inserted into the chamber, and the disk was rotated simultaneously to obtain the normal

$U_3Si_2$  powder. There were no unusual findings in the ceramic crucible, but the molten alloy fell to the periphery of the disk instead of the center. This led to an unstable powder spraying process, resulting in  $U_3Si_2$  powder adhering to surrounding components and causing loss as shown in Figure 5. The yield for  $U_3Si_2$  powder with diameters less than  $150\ \mu m$  was 29.4%, which represents an improvement compared to the first experiment (C2405 batch).

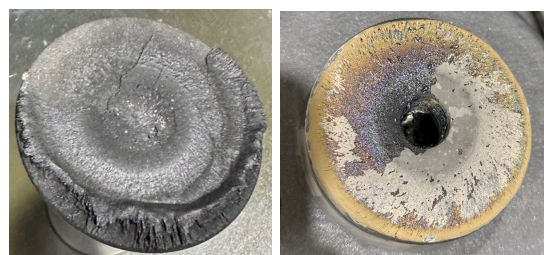


Fig. 5. Atomization Component Parts after third experiment (C2407 batch) : Disk(Left), Carbon Parts(Right)



Fig. 6.  $U_3Si_2$  powder of C2407 batch before sieving

In the fourth experiment (C2408 batch), it was confirmed that there were no leaks from the molten alloy up to  $2050^\circ C$ . For safety reasons, this experiment was terminated without proceeding with any further spraying and powder fabrication processes. As shown in Figure 7, the entire  $U_3Si_2$  alloy ingot remained intact in the ceramic crucible.



Fig. 7.  $U_3Si_2$  alloy ingot in the ceramic crucible after fourth experiment (C2408 batch)

The results of the yield rate for  $U_3Si_2$  powder with diameters less than  $150\ \mu m$  over the experiments (C2405~C2408 batches) were summarized in Table II.

Table II: Non-rod Experiment's yield rate for  $U_3Si_2$  powder with diameters less than  $150\ \mu m$

Batch Name	Yield rate (%)	Remark
C2405	15.8	-
C2406	-	Failure related to applying coating agents
C2407	29.4	-
C2408	-	Stopped on purpose before spraying

### 3. Conclusion

The experiments demonstrated that by applying a specific coating method, it is possible to control automatic outflow and fine trickling of the molten  $U_3Si_2$  alloy through the ceramic crucible's nozzle without using the pouring rod at high temperatures (near  $2000^\circ C$ ). However, to fully implement the advanced non-rod atomization technology, further experiments are needed to address the following issues:

1. Increase the loading capacity to more than 3 kg.
2. Adjust the control of the molten  $U_3Si_2$  alloy to fall at the center of the disk to ensure proper spraying.
3. Improve the yield rate for  $U_3Si_2$  powder with diameters less than  $150\ \mu m$  to over 90%.

Once these improvements are completed and applied to the atomization process, many problems caused by the pouring rod will be resolved. These achievements are expected to significantly contribute to the stable fabrication for atomized  $U_3Si_2$  powder at KAERI.

### REFERENCES

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